

## A Geometrical Notation for Stacking Faults in Close-Packed Structures

BY DHANANJAI PANDEY

*School of Materials Science and Technology, Banaras Hindu University, Varanasi-221005, India*

(Received 18 January 1983; accepted 30 April 1984)

### Abstract

A new geometrical notation based on the use of intrinsic-extrinsic classification in conjunction with the concept of assigning subscripts to close-packed layers with different environments is proposed. The terms intrinsic and extrinsic have been used in a purely geometrical sense and are not linked with specific formation processes for stacking faults. It is shown that the proposed notation can provide a very compact and unique way of representing fault configurations in all close-packed structures, including the long-period polytypes.

Several different notations\* have been used for representing stacking faults in close-packed structures, none of which can provide a compact and unambiguous notation (see Pandey & Krishna, 1981, 1982). An alternative notation, which is based on the use of intrinsic-extrinsic classification (Frank, 1951) in conjunction with the concept of assigning subscripts to different close-packed layers (Prasad & Lele, 1971; Pandey & Krishna, 1976), is proposed. In this notation, the terms intrinsic and extrinsic have been used in a purely geometrical sense in accordance with the original suggestion of Frank and not by associating these terms with specific processes of formation like condensation of a sheet of vacancies and interstitials respectively. As shown below, this notation can provide a very compact and unique way of representing fault configurations in all close-packed structures including long-period structures observed in polytypic materials. We shall develop this notation in relation to the possible fault configurations in a hexagonal structure, namely  $6H\ ABCACB$ , which is the most common high-temperature modification of SiC. An application of this notation to a rhombohedral structure, namely  $9R$ , is also presented.

\* For a detailed account of the various notations, the reader is referred to the paper *Stacking Faults in Close-Packed Structures: Definitions and Notations* (Contents: Introduction; Existing Notations: Growth and Deformation Faults, Extrinsic and Intrinsic Faults, The ' $\Delta$ - $\nabla$ ' Notation, The ' $h$ - $k$ ' Notation; Present Notation; Application of the Notation; Conclusion) by D. Pandey. This paper has been deposited with the British Library Lending Division as Supplementary Publication No. SUP 39176 (23 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

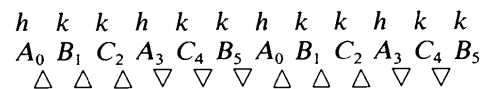
Table 1. Possible fault configurations in the  $6H$  ( $A_0B_1C_2A_3C_4B_5, \dots$ ) structure

Fault configuration ABC sequence		Subscript notation
... A B C A C B A <sub>0</sub> C <sub>0</sub>	A B C B A C ...	$I_{0,0}$
... A B C A C B A <sub>0</sub> C <sub>1</sub>	A B A C B C ...	$I_{0,1}$
... A B C A C B A <sub>0</sub> C <sub>2</sub>	A C B A B C ...	$I_{0,2}$
... A B C A C B A <sub>0</sub> C <sub>3</sub>	B A C A B C ...	$I_{0,3}$
... A B C A C B A <sub>0</sub> C <sub>4</sub>	B A B C A C ...	$I_{0,4}$
... A B C A C B A <sub>0</sub> C <sub>5</sub>	B C A B A C ...	$I_{0,5}$
... A B C A C B A <sub>1</sub> B <sub>0</sub>	B C A C B A ...	$I_{1,0}$
... A B C A C B A <sub>1</sub> B <sub>1</sub>	B C B A C A ...	$I_{1,1}$
... A B C A C B A <sub>1</sub> B <sub>2</sub>	B C C B A C ...	$I_{1,2}$
... A B C A C B A <sub>1</sub> B <sub>3</sub>	C B A B C A ...	$I_{1,3}$
... A B C A C B A <sub>1</sub> B <sub>4</sub>	C B C A B A ...	$I_{1,4}$
... A B C A C B A <sub>1</sub> B <sub>5</sub>	C A C B A A ...	$I_{1,5}$
... A B C A C B A <sub>2</sub> B <sub>0</sub>	C A B A C B ...	$I_{2,0}$
... A B C A C B A <sub>2</sub> B <sub>1</sub>	C A C B A B ...	$I_{2,1}$
... A B C A C B A <sub>2</sub> B <sub>2</sub>	C B A C A B ...	$I_{2,2}$
... A B C A C B A <sub>2</sub> B <sub>3</sub>	A C B C A B ...	$I_{2,3}$
... A B C A C B A <sub>2</sub> B <sub>4</sub>	A C A B C B ...	$I_{2,4}$
... A B C A C B A <sub>2</sub> B <sub>5</sub>	A B C A C B ...	$I_{2,5}$
... A B C A C B A <sub>3</sub> C <sub>0</sub>	B C A C B A ...	$E_0$
... A B C A C B A <sub>3</sub> C <sub>1</sub>	A C A C B A ...	$E_1$
... A B C A C B A <sub>3</sub> C <sub>2</sub>	B A C B A B C ...	$E_2$

Note: dotted vertical lines represent the location of the fault plane with respect to the initial stacking sequence on the left-hand side.

Enantiomorphous pairs:  $I_{0,1}$  &  $I_{2,3}$ ,  $I_{0,2}$  &  $I_{1,3}$ ,  $I_{1,1}$  &  $I_{2,2}$ ,  $I_{1,4}$  &  $I_{2,5}$ , and  $E_0$  &  $E_2$ .

The  $6H$  ( $ABCACB, \dots$ ) structure can also be written as  $hkkhkk, \dots$  or as  $\Delta\Delta\Delta\nabla\nabla\nabla, \dots$  in the ' $h$ - $k$ ' and ' $\Delta$ - $\nabla$ ' notations (for details see Pandey & Krishna, 1982). Six kinds of layers need to be distinguished in the perfect  $6H$  structure of which three layers are in the  $\Delta$  orientation with respect to the preceding layers and the remaining three in the  $\nabla$  orientation. Let us choose the origin on an ' $h$ ' type layer which is followed by two ' $k$ ' layers such that the layer next to the origin is in the  $\Delta$  orientation. Let the layer through the origin be denoted by a subscript 0 and succeeding layers in the unit cell by subscripts 1, 2, 3, 4 and 5. Then the perfect  $6H$  structure can be written as



There are six ways in which layers in the unit cell of the  $6H$  structure can be written with the first layer in  $A$  orientation: (i)  $A_0B_1C_2A_3C_4B_5$ , (ii)  $A_1B_2C_3B_4A_5C_0$ , (iii)  $A_2B_3A_4C_5B_0C_1$ , (iv)  $A_3C_4B_5A_0B_1C_2$ , (v)  $A_4C_5B_0C_1A_2B_3$ , (vi)  $A_5C_0A_1B_2C_3B_4$ . Similarly there are six ways in which the structure of  $6H$  can be written starting with the



the fault configurations  $I_{0,\bar{0}}$ ,  $I_{0,\bar{1}}$ ,  $I_{0,\bar{2}}$  and  $I_{1,2}$  are enantiomorphous with  $I_{1,\bar{1}}$ ,  $I_{1,\bar{2}}$ ,  $I_{2,\bar{1}}$  and  $I_{2,0}$  respectively, leaving only 14 unique intrinsic fault configurations in the  $9R$  structure.

The author is grateful to Dr S. Lele for critically reading the manuscript.

*Acta Cryst.* (1984). **B40**, 569–574

## Cell Volumes of $AP\text{O}_4$ , $AV\text{O}_4$ , and $AN\text{bO}_4$ Compounds, where $A = \text{Sc, Y, La-Lu}^*$

BY A. T. ALDRED

*Materials Science and Technology Division, Argonne National Laboratory, Argonne, IL 60439, USA*

(Received 31 October 1983; accepted 30 May 1984)

### Abstract

Precision lattice constants have been determined for a series of  $AP\text{O}_4$ ,  $AV\text{O}_4$ , and  $AN\text{bO}_4$  compounds (where  $A$  is a rare-earth ion, Sc, Y, or La) with the monazite, zircon, and fergusonite structures. The linear variation of the cube root of the unit-cell volume with the tabulated radius of the  $A$  ion [Shannon (1976). *Acta Cryst.* **A32**, 751–767] has been evaluated. Within the rare-earth series, only the Gd compounds have cell volumes which deviate from straight-line behavior, a result which implies that  $r_{\text{Gd}}$  (CN=8) should be  $1.196(1) \text{ \AA}$ . Systematic discrepancies are also noted for the corresponding Sc, Y, and La compounds which indicate that in these compounds the modified values of the radii should be  $r_{\text{Sc}} = 1.020(2)$ ,  $r_{\text{Y}} = 1.154(1)$ , and  $r_{\text{La}} = 1.313(2) \text{ \AA}$ . Literature cell constants have been used, in conjunction with the above analysis, to obtain radii for the analogous trivalent actinide ions,  $r_{\text{Pu}} = 1.250(5)$ ,  $r_{\text{Am}} = 1.237(5)$ , and  $r_{\text{Cm}} = 1.224(10) \text{ \AA}$ .

### Introduction

The concept of an ionic, or crystal, radius is still a most useful one in understanding the structures and stabilities of complex oxide phases. Shannon & Prewitt (1969) and Shannon (1976) have been able to develop, in part by an analysis of experimentally determined lattice constants and unit-cell volumes, a systematic series of empirical radii for different valence and coordination states. Muller & Roy (1974) have related the occurrence of specific crystal structures in complex oxide phases ( $A_xB_yO_z$ ) to these radii by use of the structure-field map, where a stability

### References

- FRANK, F. C. (1951). *Philos. Mag.* **42**, 809–819.  
 PANDEY, D. & KRISHNA, P. (1976). *Acta Cryst.* **A32**, 488–492.  
 PANDEY, D. & KRISHNA, P. (1981). *Ind. J. Pure Appl. Phys.* **19**, 796–802.  
 PANDEY, D. & KRISHNA, P. (1982). In *Current Topics in Materials Science*, Vol. 9, edited by E. KALDIS. Amsterdam: North-Holland.  
 PRASAD, B. & LELE, S. (1971). *Acta Cryst.* **A27**, 54–64.

diagram is produced in which the coordinates of the phase space are the radii of the  $A$  and  $B$  ions.

As an initial phase of a program to focus in more detail on the concept of a structure-field map, precision lattice parameters have been determined for a series of  $A^{3+}P^{5+}O_4$ ,  $A^{3+}V^{5+}O_4$ , and  $A^{3+}Nb^{5+}O_4$  compounds where  $A$  is a rare-earth ion, scandium, yttrium, or lanthanum. Unit-cell volumes obtained from these lattice parameters have been analyzed in terms of the size of the  $A$  ion, and the results indicate that the radii of some of the  $A$  ions in these compounds may be different from the listed values (Shannon, 1976). Additional analysis of the unit-cell volumes of the analogous trivalent actinide compounds yields the corresponding radii of Pu, Am, and Cm in eightfold coordination.

The compounds used in this study were chosen, in part, because the structure-field map given by Muller & Roy (1974, Fig. 18) demonstrates a powerful systematization of the existing data, and because of the chemical and structural similarities between the phases that occur in this family of compounds. The monazite structure, which occurs primarily among the early lanthanide orthophosphates (prototype  $\text{CePO}_4$ ), is monoclinic, space group  $P2_1/n$ . The zircon structure (prototype  $\text{ZrSiO}_4$ ) is tetragonal, space group  $I4_1/amd$ , and occurs widely among the heavier lanthanide orthophosphates and arsenates, the lanthanide orthovanadates, and some actinide orthosilicates. The fergusonite structure (prototype  $\text{YNbO}_4$ ) is monoclinic, space group  $I2/c$ , and occurs among the lanthanide (and actinide) orthoniobates and tantalates. Each of the three structure types contains isolated  $\text{BO}_4$  tetrahedra interlinked by  $A$  ions which have primarily eightfold oxygen coordination. The structural and chemical similarities have been discussed elsewhere (Aldred, 1984).

\* Work supported by the US Department of Energy.